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# Synthesis of SiO<sub>2</sub>@Ag Nanocomposite for Investigating Metal-Enhanced Fluorescence and Surface-Enhanced Raman Spectroscopy

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#### Abstract

 $SiO_2@Ag$  nanocomposite (NC) has been synthesized by the chemical reduction and Stöber method for Metal-enhanced fluorescence (MEF) of Rhodmine 6G (R6G) and Surface-enhanced Raman spectroscopy (SERS) of Malachite green (MG). As-synthesized SiO\_2@Ag NC indicated SiO\_ nanosphere (NS) and Ag nanoparticle (NP) morphologies. The SiO\_2@Ag NC was high quality with a well-defined crystallite phase with average sizes of 24 nm and 132 nm for Ag NP and SiO\_ NC, respectively. By using SiO\_2@Ag NC, the photoluminescence (PL) intensity of the R6G (at 59.17 ppm) was increased approximately 133 times. The SERS of the MG (at 1.0 ppm) with SiO\_2@Ag NC as substrate clearly observed vibrational modes in MG dye at 798, 916, 1172, 1394, and 1616 cm<sup>-1</sup>. As a result, the SERS enhancement factor (EF<sub>SERS</sub>) at 1172 cm<sup>-1</sup> obtained 6.3 × 10<sup>6</sup>. This initial study points to the potential of SiO\_2@Ag NC as a promising material for MEF and SERS substrates to detect dyes at low concentrations.

Keywords SiO2@Ag · Nanocomposite · Rhodamine 6G · SERS · Malachite green · Enhancement factor

#### Introduction

Nowadays,  $SiO_2$ -combined Ag NPs have been widely applied in bio-medicine [1, 2], aesthetics [3], optoelectronics [4, 5], pharmaceutical medicine [6], food technology [7, 8], early diagnosis [9], disease treatment [10], and environmental treatment [11–13]. The SiO<sub>2</sub> ingredient possesses

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thermal-stable, chemical-inert, and non-toxic properties. In addition, the SiO<sub>2</sub> nanostructure has been easily prepared in nanoparticle form and dispersed evenly in various solutions. Due to the high biocompatibility and functionalized surface, SiO<sub>2</sub> NPs can easily bond with pigments, metal ions, and biomolecules [11]. While Ag NPs- the rest of the NCs possess the surface plasmon resonance (SPR) peak at about 400 nm can produce SERS and MEF signals larger than that of metals with the absorption peak at visible or near-infrared regions [11, 14]. In the electromagnetic field of the incident light, the gap positions between Ag NPs are also known as hot spots, so the local electromagnetic field has been enhanced [11, 14, 15]. The intensity of the electromagnetic field gradually decreases from the center of the hot spots away.

SiO<sub>2</sub>@Ag NC structures have been successfully prepared using different methods, which can control the morphology and size of SiO<sub>2</sub>@Ag NCs, thereby affecting the MEF and SERS capabilities [16–20]. For example, SiO<sub>2</sub>@ Ag nanostructures were synthesized by sol–gel method with the decorated Ag NPs size of 35 nm on nano-sized SiO<sub>2</sub> spheres of 20–60 nm [16]. The co-precipitation process combined with polymerization created Ag@SiO<sub>2</sub> NCs with the spherical Ag core size in the order of micrometers within the SiO<sub>2</sub> shell [17]. Using the Stöber method, synthesized Ag@SiO<sub>2</sub> nanospheres had 80 nm sized Ag NPs on 800 nm sized SiO<sub>2</sub> spheres [21], core–shell Ag@SiO<sub>2</sub> NCs were composed of 50 nm sized Ag cores and 75 nm sized SiO<sub>2</sub> shells [22].

R6G is an organic dye widely used in industrial products for daily life. However, the release of the R6G dye into the environment will cause pollution and affect human health. R6G molecules always tend to form dimers or monomers. A charged redistribution in the molecules changes their fluorescent property [23–26]. PL spectrum often appears as a wide band with a peak position in the range of 550-630 nm, which is the superposition of some bands [27]. PL intensity and maximum position depend on the R6G concentration in the solution. At low concentrations, R6G molecules are not agglomerated into big clusters, so PL is characterized by monomers, while at more significant concentrations, it is dominated by dimers [23]. In the electromagnetic field of incident radiation, R6G molecules at the hot spots between Ag NPs receive substantial PL enhancement because of the MEF effect [28, 29]. Several studies have also shown that MEF occurs for R6G [28, 29] and other organic compounds [22]. Through supporting Ag nanoparticles, the MEF enhancement factor (EF<sub>MEF</sub>) can be increased many times [30-35].

Besides, MG dye is also studied in this study. It is popular in the dye industry and aquaculture. The MG dye can be recognized at dilute concentration via the SERS [36]. For our initial studies, MG has been used as a probe to check whether a material could be used in SERS. The SERS of the MG appears clearly with characteristic peaks for vibration of binding groups in the molecular, and the intensity depends on the interaction between MG and metal NCs. The characteristic peaks of ring C–H in-plane bending and ring C–C stretching vibrations are usually very obvious and are chosen to calculate the enhancement factor [37, 38]. Enhancement factor determination allows it to appreciate the enhancement level of the SERS signal. Raman signals at low concentrations can reach approximately  $10^5 - 10^8$  thanks to the SERS of Ag NPs [37, 39-42]. In fact, the SERS is used as a tool to detect the concentration of dye compounds in inspection samples. Another concern is that the facile solubility of both R6G and MG makes it spread in water sources and penetrate living organisms through the food chain [43]. Therefore, the influence of these dyes on human health has also been studied in many other works, thereby detecting the dyes, treating emitted residues, and making recommendations.

In this work, SiO<sub>2</sub>@Ag nanocomposite was synthesized through the principles of the Stöber method with some modification technologies. As-synthesized SiO<sub>2</sub>@Ag NCs would be investigated in terms of their optical and structural characteristics. At present, this SiO<sub>2</sub>@Ag NC with multitasking can be used in MEF and SERS to detect organic dyes at dilute concentrations of MG and R6G.

#### Experimental

Tetraethoxysilane (TEOS) Si(OC<sub>2</sub>H<sub>5</sub>)<sub>4</sub> (99,98%), (3-aminopropyl) trimethoxysilane (APTMS) H<sub>2</sub>N(CH<sub>2</sub>)<sub>3</sub>Si(OCH<sub>3</sub>)<sub>3</sub> (99,96%), silver nitrate AgNO<sub>3</sub> (99,99%), sodium borohydride (NaBH<sub>4</sub>), ammoniac NH<sub>3</sub> ( 25%), Rhodamine 6G (R6G), Malachite green (MG) (98%, solid, refractive index n = 1.5940, density  $\rho$  = 1.0448 g/cm<sup>3</sup>, and molecular weight M = 364.911 g), absolute ethanol were purchased from Merck. All used chemicals were analytical grade. Deionized (DI) water was used in all experiments.

For fabricating  $SiO_2@Ag NCs$ , the processe is illustrated in Scheme 1 with two steps. Firstly,  $SiO_2$  particles were obtained according to the Stöber method [19] with slight modification. 5 ml of TEOS was dissolved in 30 ml of the ethanol and DI water solvent with a volume rate of 1:1, and then this solution was ultrasonicated for 15 min at room temperature to obtain an A solution. 3 ml of NH<sub>3</sub> was added into the A solution and stirred for 4 h to obtain a B mixture. The B mixture was centrifuged and filtered by absolute ethanol and DI water three times, then dried as precipitation at 80 °C for 15 h to obtain SiO<sub>2</sub> powder.

Secondly, 0.18 g of AgNO<sub>3</sub> was dissolved in 10 ml of DI water and 3 ml of NH<sub>3</sub> solution by magnetic stirring to make the C solution. Similarly, 0.01 g of NaBH<sub>4</sub> was dissolved in 10 ml of DI water to make the D solution. 0.2 g of assynthesized SiO<sub>2</sub> was ultrasonicated with 50 ml ethanol for 15 min at room temperature to make an E solution. 0.1 ml of APTMS was added to the E solution and stirred for 15 h to produce the SiO<sub>2</sub>-APTMS mixture (F). Then, the F mixture was centrifuged to obtain SiO<sub>2</sub>-APTMS precipitations. The SiO<sub>2</sub>-APTMS precipitations were added to the C solution and stirred for 2 h to make the G mixture. After that, the D solution was added to the G mixture and stirred for 15 min to obtain SiO<sub>2</sub>@Ag precipitations. The precipitations were washed thrice with DI water and ethanol, then dried at 80 °C for 15 h to produce SiO<sub>2</sub>@Ag NCs. The formation processes of the SiO<sub>2</sub>@Ag nanocomposite are expressed in (1-5) as follows [2]:

$$Si(OC_2H_5)_4 + 2NH_4OH \rightarrow SiO_2 + 4C_2H_5OH + 2NH_3 \quad (1)$$

$$AgNO_3 + 3NH_4OH \rightarrow [Ag(NH_3)_2]OH + NH_4NO_3 + 2H_2O$$
(2)

 $SiO_2 + APTMS + [Ag(NH_3)_2]OH \rightarrow SiO_2 - APTMS@[Ag(NH_3)_2]OH$ (3)

$$NaBH_4 + 4H_2O \rightarrow NaB(OH)_4 + 4H_2$$
(4)





$$2[Ag(NH_3)_2]OH + 4NaB(OH)_4 + 6H_2O \rightarrow 2Ag^{\circ} + 15H_2 + 4H_3BO_3 + 4NaNO_3$$
(5)

The synthesis process of  $SiO_2@Ag$  is presented briefly in Step a of the Scheme 1.

Thirdly, in order to investigate the MEF of SiO<sub>2</sub>@Ag NCs for the R6G dye, SiO<sub>2</sub>@Ag NCs and R6G were dispersed into DI water to make two different solutions at concentrations of 35.5 and 355 ppm, respectively. Two above solutions were mixed together to get solutions with the SiO<sub>2</sub>@ Ag to R6G volume ratio of (0:1), (1:1), (2:1), (3:1), (4:1), (5:1), (6:1), and (7:1), containing the R6G concentrations of 355, 177.5, 118.33, 88.75, 71, 59.17, 50.71, and 44.38 ppm, respectively. An equal amount of each solution was taken with a micropipette, deposited on the Si substrate, and let dry naturally. As-deposited Si substrates, samples, were denoted as R(0:1), R(1:1), R(2:1), R(3:1), R(4:1), R(5:1), R(6:1), and R(7:1) for MEF investigation.

Finally, with respect to preparations for the study of the SERS, the Si substrate with edge sizes of  $(1.0 \text{ cm} \times 1.0 \text{ cm})$  was cleaned in acetone, ethanol, and DI water, soaking in 50% HNO<sub>3</sub> for 15 min and in HF for 15 min. Let them dry naturally to get as-treated Si substrates. Then, a quantity of SiO<sub>2</sub>@Ag solution with the 35.5 ppm concentration was deposited on the above as-treated Si substrates and dried to get SERS substrates. Afterwards, 10 µl of the 1.0 ppm MG solution was deposited on the SERS substrate area of 12.46 mm<sup>2</sup>, drying to get the 1.0 ppm MG SERS substrate. MG with concentrations of  $10^3$  and  $10^6$  ppm were also deposited on Si substrate. As a result, there were four types of substrates: SiO<sub>2</sub>@Ag,  $10^3$  ppm MG, and  $10^6$  ppm MG on Si substrates and 1.0 ppm MG on SERS substrate, which were

used for the SERS research, labelled as  $SiO_2@Ag$ ,  $10^3$  ppm MG,  $10^6$  ppm MG, and 1.0 ppm MG, respectively. The preparation of substrates for the study of MEF and SERS is shown in Step b of the Scheme 1.

The microstructure and morphology of SiO<sub>2</sub> and SiO<sub>2</sub>@ Ag NC were investigated by X-ray diffraction (XRD) on a PANalytical Empyrean device using Cu\_K<sub>a</sub> radiation  $(\lambda = 1.54056 \text{ Å}, 2\theta = 15^{\circ}-70^{\circ})$  and high-resolution transmission electron microscopy JEM-2100. The absorption spectra were recorded on the UV-2450 system. PL spectra were excited by the 325 nm radiation of the He-Cd laser on a Spectra Pro2500i system. Raman spectra were recorded on a portable  $\mu$  Raman-Ci (Technospex) spectroscopy using a diode laser with technical specifications of power (10 mW), an excitation radiation (785 nm), a numerical aperture (NA=0.5), and a laser spot area (2.9  $\mu$ m<sup>2</sup>). All spectra were recorded at room temperature.

#### **Results and Discussions**

#### Structure and Morphology of SiO<sub>2</sub>@Ag Nanocomposite

The crystalline structure phase of SiO<sub>2</sub> and SiO<sub>2</sub>@Ag NCs is shown by XRD patterns in Fig. 1. For the SiO<sub>2</sub>, a broadband occurred at about 23.91° with weak intensity characterized by an amorphous structure phase. For the SiO<sub>2</sub>@Ag NCs, the diffraction peaks at 37.94, 44.28, and 64.38° attributed to (111), (200), and (220) atomic planes of a face-centered cubic structure phase of Ag NPs, respectively, according to PDF Nº: 96–901-3047, which belongs to Fm3m space group. The diffraction peak of the (111) plane is narrow and most robust in the others; it is predicted that Ag particles achieve high crystallinity and big crystal sizes. While the characteristic band of the SiO<sub>2</sub> structure phase at 23.91° is weak, it further highlights the presence of Ag particles. From the XRD pattern, using Debye Scherrer and Bragg's formulas [3, 4, 16], the average particle size and lattice d-spacing of Ag NPs are calculated at about 24 nm and 2.38 Å, respectively. The characteristic peaks for the silver structural phase in the XRD pattern of SiO<sub>2</sub>@ Ag NCs showed that Ag NPs had been formed on the SiO<sub>2</sub> surface [21]. For easy visualization, the mechanism of Ag NP decorated on SiO<sub>2</sub> NSs is described as follows: the F solution is a SiO<sub>2</sub>-APTMS colloidal system containing negative NH<sub>2</sub>amin groups. When adding the  $Ag[(NH_3)]_2^+$  ion solution to the F, Ag<sup>+</sup> ions have been absorbed on the SiO<sub>2</sub> surface due to the interaction between NH2- groups and Ag(NH3)+ ions to form the G mixture  $(SiO_2-APTMS-Ag[(NH_3)]_2^+$ . Therefore, after the NaBH<sub>4</sub> solution was added, Ag<sup>+</sup> ions on the surface of SiO<sub>2</sub> NSs were reduced to Ag NPs. Those are illustrated in Fig. 2 [44]. The structural phase, shape, and size of the SiO<sub>2</sub>@Ag NCs are shown in Fig. 3. The TEM image showed



Fig. 1 XRD patterns of SiO<sub>2</sub> and SiO<sub>2</sub>@Ag NCs

that SiO<sub>2</sub> particles possessed spherical shape with sizes in the range of 100–180 nm (Fig. 3a), but the distributed SiO<sub>2</sub> NSs concentrated at an average size of 132 nm (Fig. 3c). The inset showed that the Ag NPs on SiO<sub>2</sub> spheres were not agglomerated. HR-TEM image is observed that the lattice ridges are oriented in the same parallel direction representing a single grain (Fig. 3c). The d-spacing between two consecutive (111) planes was determined about 2.4 Å (the inset of Fig. 3c) likely an estimated result by the XRD pattern [13]. It also confirmed that Ag NPs were decorated on the SiO<sub>2</sub> NSs. The SAED pattern of Ag NP consists of bright spots representing the crystal structure phase (Fig. 3d) [5]. Basically, the bright spots are located on the concentric circle fringes, in which three fringes correspond to the (111), (200), and (220) planes in the XRD pattern [9].

# The MEF and SERS Properties of SiO<sub>2</sub>@Ag Nanocomposites

Figure 4 presents the UV–Vis absorption spectrum (a) of  $SiO_2@Ag$  NCs and R6G solutions and PL spectra (b) of  $SiO_2@Ag$  NCs and R6G solutions deposited on Si substrate with the concentration of 35.5 and 355 ppm (R(1:0), R(0:1)), respectively. The Gaussian symmetry of the UV–Vis absorption spectrum of  $SiO_2@Ag$  evidenced that Ag NPs have a quasi-spherical shape with a concentrated size distribution [16]. The maximum position of that absorption spectrum is centered at about 400 nm, derived from the SPR of Ag NPs



(Fig. 4a) [14]. The SPR spectrum is only recorded on the metal surface, proving that the reduced Ag NPs are attached to the SiO<sub>2</sub> surface. There are several factors responsible for the enhancement. The first is the effect of local field enhancement generated near metallic structures [45, 46]. Another factor is the plasmon-coupling effect mediated by a non-radiative interaction [47]. If the plasmonic structure and the fluorophore are at an optimal distance, the energy transfer between them is dominated. Those are explained by Förster (or fluorescence) resonance energy transfer, the mechanism of electron transfer through molecules. The

non-radiative energy transfer between metal and fluorophore depends not only on the strength of the electric field but also on the degree of spectral overlap between the metal surface and the fluorophore [46]. A broad 525 nm absorption band appeared in the UV- Vis absorption spectrum of R6G (Fig. 4a). There is quite a good overlap between the absorption of R6G and SiO<sub>2</sub>@Ag.

Therefore, R6G is used to study the ability of fluorescence enhancement of SiO<sub>2</sub>@Ag NCs. In the R6G molecular structure, there are  $\pi$ -bonds of C=C, C=N, and C=O groups, as seen in the inset of Fig. 4a. The optical properties



**Fig. 3** a TEM image, **b** particle distribution histogram, **c** HR-TEM image, and **d** SAED pattern of SiO<sub>2</sub>@Ag NCs **Fig. 4 a** UV–Vis absorption spectra of SiO<sub>2</sub>@Ag NCs and R6G solutions and inserted figure as R6G molecule, and **b** PL spectra of SiO<sub>2</sub>@Ag NCs and R6G on Si substrates



of the R6G are related to the electronic transitions between  $\pi$  and  $\pi^*$  orbital states [48]. As observed in the PL spectrum of SiO<sub>2</sub>@Ag in Fig. 4b, the broadband at about 430 nm originated from different defect centers of SiO<sub>2</sub> NSs [49, 50]. The PL spectrum of the R6G appears to be a band at 600 nm. However, it is robust in comparison with SiO<sub>2</sub>@Ag in terms of the PL intensity. This band is caused by the emission of dimers [23–26], but the nature is still related to the electric transitions from S<sub>1</sub>( $\pi^*$ ) excited state to S<sub>0</sub>(n,  $\pi$ ) ground state [48]. Furthermore, the R6G concentration of 355 ppm seems quite dense, as the PL maximum peak is located at 600 nm, belonging to the sort of dimer [23].

Notably, there is a strong enhancement in both MEF and SERS using SiO<sub>2</sub>@Ag NCs. Figure 5 manifests the PL spectra of the R6G with different volume ratios between SiO<sub>2</sub>@Ag and R6G, in which the PL spectrum of R6G is redrawn. All PL spectra show a band with a maximum in the range of 553–560 nm but not at 600 nm, as explained in Fig. 4b. The shape of PL spectra is asymmetric, left-steep, and right-gentle due to the overlap of dimer-monomer PL bands, herein, the monomer band is more dominant than the dimer one [27, 51]. In the case of a SiO<sub>2</sub>@Ag NCs absence, the PL spectrum of the R6G (R(0:1)) appears extremely weak, resembling a baseline. However, using SiO<sub>2</sub>@Ag NCs, the PL intensity of the R6G becomes much stronger, and the peak position of the PL band depends on the change of the SiO<sub>2</sub>@Ag to R6G volume ratio. In order to estimate the  $EF_{MEF}$ , a compared calculation between the absolute PL intensity of the R6G at R(5:1) and the one at R(0:1) was expressed as [28, 29]:

$$EF_{MEF} = (I_{R(5:1)} - I_{SiO_2@Ag}) \cdot (I_{R(0:1)} - I_{SiO_2@Ag})^{-1}$$
(6)

In Eq. (6),  $I_{R(5:1)}$ ,  $I_{(R(0:1))}$ , and  $I_{SiO_2@Ag}$  are absolute intensities according to the SiO<sub>2</sub>@Ag to R6G volume ratios and SiO<sub>2</sub>@Ag substrate. Note that the spectral signal from the Si substrate was feeble in the observed region under the excitation radiation of 325 nm; it was not mentioned. When the volume ratio is R(1:1), the absolute PL intensity of the 560 nm band is 24 times as much as that of the 600 nm band

**Fig. 5** a PL spectra of R6G using  $SiO_2@Ag$  NCs, and b Peak position and enhancement factor versus  $SiO_2@Ag$  to R6G volume ratios



of R(0:1). From R(2:1) to R(4:1), the PL intensity also goes up while the maximum position shifts towards to the shorter wavelength. At the volume ratio of R(5:1), the PL intensity reaches the maximum and rises by 133 times in comparison with the intensity of R(0:1). Afterwards, the PL intensity decreases despite increasing the volume ratio from R(6:1)to R(7:1). The maximum position and  $EF_{MFF}$  as functions of volume ratios are delineated in Fig. 5b. The change in optical R6G spectra according to the SiO<sub>2</sub>@Ag to R6G volume ratio can be explained, based on the distance change between R6G and Ag NPs. It is known that a suitable distance between R6G and Ag NPs will exist for the best energy transfer process so that the PL intensity is the strongest. Yet the PL intensity will be quenched if the formed distance differs from the above suitable one [25, 52]. In the case of R(5:1), about 59.17 ppm of the R6G, the distance between R6G and Ag NPs can be the most suitable for the energy transfer, increasing the PL [25]. In addition, it can also be due to the change of SiO<sub>2</sub>@Ag to R6G volume ratio, as the R6G concentration is changed, leading to the change of the energy transfer distance [53]. The energy transfer process between coated polymers and nanoparticles of composite materials has also been known [54, 55]. It can be seen that, herein, the surface-functionalized SiO<sub>2</sub> spheres possessed the functional -NH<sub>2</sub> groups of APTMS molecules as anchors attracting both Ag NPs and R6G molecules on the surface of the SiO<sub>2</sub> NSs, so there always ensured a suitable distance between them for the MEF effect. Therefore, Ag NPs decorated SiO<sub>2</sub> NSs also have a good MEF in comparison with Ag NPs [29–31, 33], coated Ag NPs [28, 32], core–shell Ag@SiO<sub>2</sub> structures [22, 34], and Ag multilayers [35], as shown in Table 1.

As-synthesized SiO<sub>2</sub>@Ag NCs have not only the ability of MEF but also SERS. Figure 6 presents Raman spectra of SiO<sub>2</sub>@Ag,  $10^3$ ,  $10^6$ , and 1.0 ppm MG on different substrates. In the case of SERS (at 1.0 ppm MG), a series of peaks appear clearly at 798, 916, 1172, 1365, 1394, and 1613 cm<sup>-1</sup> which are attributed to vibrations, respectively, ring C-H out-of-plane bending (798 and 916 cm<sup>-1</sup>), ring C-H in-plane bending (1172 cm<sup>-1</sup>), N-phenyl stretching  $(1365 \text{ cm}^{-1})$ , C-N symmetrical stretching  $(1394 \text{ cm}^{-1})$ , and ring C–C stretching  $(1613 \text{ cm}^{-1})$  [36, 56]. While the Raman signals of SiO<sub>2</sub>@Ag and 10<sup>3</sup> ppm MG are recorded too weak, peaks are not shown obviously in Fig. 6a. It is affirmed that the peaks at 798, 916, 1172, 1365, 1394, and  $1613 \text{ cm}^{-1}$  do not belong to the SiO<sub>2</sub>@Ag and Si substrates. The SERS was not caused by the Si substrate, even though the MG concentration was utilized to be  $10^3$  ppm [57], or owing to the characteristic peak of the Si substrate in different wavenumber regions [58]. Only if the MG concentration is equal to  $10^6$  ppm, a series of peaks at 1172, 1394, and 1613 cm<sup>-1</sup> are observed in the normal Raman spectrum. As seen in the  $1172 \text{ cm}^{-1}$  mode in Fig. 6b, the intensity of the SERS signal is stronger than that of the normal Raman signal; this is also true for the others. The cause of the SERS is due to the resonance of the vibrational modes with the electromagnetic field at the hot spots, as mentioned above. To evaluate the enhancement factor of the SERS intensity on

Table 1 Comparison of EF<sub>MEF</sub> and EF<sub>SERS</sub> values of various materials for organic compounds

Materials	Organic compounds	EFMEF	EF <sub>SERS</sub>	References
Ag@SiO <sub>2</sub> @RITC@SiO <sub>2</sub>	Rhodamine B isothiocyanate	5	-	[22]
SiO <sub>2</sub> -Ag-cicada wing	R6G	1.99	-	[28]
Flowerlike silver nanostructures	R6G	71	-	[29]
Silver nanostructures	SH-oligomer-Cyanine 5	25	-	[30]
Silver islands films	Rhodamine B	6	-	[31]
Ag@SiO <sub>2</sub> NCs	Glucose	4.9	-	[32]
Silver NPs	Fluorescein, Cyanine 3, and Cyanine 5	10-20	-	[33]
Ag@SiO <sub>2</sub> NCs	Rhodamine 800	20	-	[34]
Silver-SiO <sub>2</sub> -Silver Nanoburger	Fluorescein isothiocyanate	35		[35]
SiO <sub>2</sub> @Ag NCs	R6G	133	-	This work
Starch-coated Ag NPs	MG	-	$5.11 \times 10^{5}$	[38]
Ag decorated microstructured PDMS	MG	-	$2.06 \times 10^{5}$	[39]
Ag NPs	MG	-	$7.9 \times 10^{6}$	[40]
Ag NPs modified glass fiber paper	MG	-	$1.3 \times 10^{8}$	[41]
Ag NPs	MG	-	$1.5 \times 10^{5}$	[42]
SiO <sub>2</sub> @Ag NCs	MG	-	$6.3 \times 10^{6}$	This work

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**Fig. 6** a Raman spectra of  $SiO_2@Ag, 10^3$ , and  $10^6$  ppm MG deposited on Si substrate and 1.0 ppm MG deposited on SERS substrate and **b** A region of enlarged Raman spectra for 1172 cm<sup>-1</sup> peak



the SiO<sub>2</sub>@Ag substrate in comparison with normal Raman intensity on the Si substrate, the  $EF_{SERS}$  can be determined by the following formula (7) [59]:

$$EF_{SERS} = (I_{SERS}.C_{Raman}) / (I_{Raman}.C_{SERS})$$
(7)

where  $I_{SERS}$  and  $I_{Raman}$  are the integrated intensities of the SERS and the normal Raman spectrum, respectively. C<sub>SERS</sub> and C<sub>Raman</sub> are the concentrations of MG solutions deposited on the SiO<sub>2</sub>@Ag substrate for SERS analysis and the Si substrate for normal Raman analysis. Note that the diode laser worked under the same experimental conditions for all SERS and normal Raman analyses. Used MG volume and MG-covered surface area on both substrates are 10 µl and 12.46 mm<sup>2</sup>, respectively, as presented in experiments. Consider that SiO<sub>2</sub>@Ag NCs and MG molecules are distributed uniformly on the surface of Si substrates. Thus, the C<sub>SERS</sub> and C<sub>Raman</sub> equal 1.0 and 10<sup>6</sup> ppm, respectively. For the 1172 cm<sup>-1</sup> peaks, by integrating the intensity, the EF<sub>SERS</sub> was estimated to be approximately  $6.3 \times 10^6$ . The calculated EF<sub>SERS</sub> value is sensitive to 1.0 ppm MG and close to the other results in Table 1 [38-42].

Thus, it can be concluded that the as-fabricated  $SiO_2$  @ Ag NCs have the capabilities for MEF and SERS. Ag NPs were glued to the spherical  $SiO_2$  surface by APTMS molecules via the reduction process, so Ag NPs are fixed on the surface of  $SiO_2$  NSs. Consequently, hot spots were also created where the dye molecules were located at a suitable distance to enhance fluorescence and Raman scattering signals. The PL intensity of the R6G increased strongly when the  $SiO_2$ @Ag content was small at the R(1:1). It reached maximum and was observed easily at the R(5:1) with the R6G concentration of 59.17 ppm. Besides, the SERS signal is obtained so strongly for probing MG dye at the concentration of 1.0 ppm.

### Conclusion

In summary, we have demonstrated a facile way to synthesize SiO<sub>2</sub>@Ag nanocomposite based on Stöber's method and a chemical reduction process. As-formed Ag nanoparticles in quasi-spherical shapes with an average crystallite size of 24 nm were decorated on the SiO<sub>2</sub> surfaces. The decoration of the Ag nanoparticle is necessary to create a suitable distance for the energy transfer between the dye and the Ag nanoparticle. The PL intensity increased by 133 times at the SiO<sub>2</sub>@Ag to R6G volume ratio of (5:1) for probing the R6G at 59.17 ppm. The SERS enhancement factor reached approximately  $6.3 \times 10^6$  to detect the MG concentration at 1.0 ppm.

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**Data Availability** The data for this work will be made available on request.

#### Declarations

Ethical Approval Not applicable.

Competing Interests The authors declare no competing interests.

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