

Synthesis of metal organic framework based on Cu and benzene-1,3,5-tricarboxylic acid (H₃BTC) by potentiodynamic method for CO₂ adsorption

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Abstract

Practical CO₂ adsorption processes require adsorbents that can be easily and quickly synthesized at a reasonable cost, while maintaining a significant adsorption performance. In this research, a metal organic framework based on Cu and H₃BTC was synthesized by the potentiodynamic method. Results showed that the nature of the solvent strongly affected the phase structure, morphology and specific surface area of the obtained material. In methanol, a mixture of 1D structure Cu-BTC (catena-triaqua μ -[1,3,5-benzenetricarboxylato]-copper (II) corresponding to the formula [Cu(BTC)(H₂O)₃]_n and micropillars structure Cu(BTC).3H₂O was obtained. The 1D structure Cu-BTC had a plate shape and a small specific surface area of 21 m²/g. In ethanol, the 3D structure Cu-BTC (Cu₃(BTC)₂) was successfully synthesized. The obtained material has a cubic shape of size from 50 nm to 500 nm and a much better specific surface area of 620 m²/g. The 3D structure Cu-BTC offered a high CO₂ adsorption capacity of 3.6 mmol/g at standard conditions, and 1.6 mmol/g at more realistic conditions.

Keywords. Cu-BTC, crystal structure, metal organic framework, CO₂ adsorption.

1. INTRODUCTION

The current concentration of CO₂ in the atmosphere, the most significant contributor to global warming, is 420 ppm, higher than about 50 % of the beginning of the industrial revolution in 1750.^[1] Meanwhile, the “acceptable” level of CO₂ concentration should be limited at below 570 ppm in 2100 to keep the rise of global temperature of less than 2 °C.^[2] To achieve this goal, big efforts should be done to reduce atmospheric CO₂ concentration. The application of new and efficient adsorbents for CO₂ capture is one of the promising solutions for this issue.^[3] In stationary point sources such as power plants or cement kilns, CO₂ emissions can be greatly reduced by using CO₂ capture technologies.^[4] In this context, the metal-organic framework (MOF) is attracting more and more attention from researchers and developers as a new group of materials for CO₂ adsorption.

MOFs may have a 1D, 2D or 3D structure and are formed from metallic ions or clusters with organic ligands. It has many remarkable properties such as high porosity, low density, high adsorption ability, catalysis ability and high specific surface area, leading to applications in various fields such as gas separation and storage, gas purification, electrocatalysis, super capacity, electrode material for battery, pollutant removal and water purification.^[5-16] MOF can be synthesized by different methods, e.g. mechanochemical method, solvothermal method, chemical method, microwave method, and electrochemical method.^[5,10,15,17-26] Electrochemical is considered a promising method because it allows synthesizing a high quantity of pure materials for a short duration in a controlled way.^[27] Besides galvanostatic or potentiostatic, potentiodynamic synthesis is mostly used to synthesize MOF thanks to the simplicity of the process.^[24]

Many factors can affect the morphology, structure and the properties of the synthesized material, such as the nature of the organic ligand or metallic ion, solvent, temperature, pH, synthesis method, and synthesis time.^[21,28-30] The research to determine suitable conditions to synthesize MOF for practical applications still attracts lots of scientists. By changing the ligands or metallic ions, the porosity of the material can be modified to obtain a new structure that allows high selective adsorption abilities. Metallic ions of the A, IB or IVB groups are often used, such as Ni, Cu, or Co, particularly Cu because it is a common metal with lots of applications.^[28,31] The most common ligands are carboxylates, phosphonates, sulfonates and N-heterocyclic compounds such as pyridine and imidazole.

Cu₃(BTC)₂ (MOF-199 or HKUST-1), formed from Cu and H₃BTC: benzene-1,3,5-tricarboxylic acid is one of the most studied MOF. It was first reported by Williams *et al.* in 1999.^[32] In 1999, Cu₃(BTC)₂ and MOF-5 were synthesized and they are two types of MOFs that were researched extensively.^[32] In 2017, the group of K. Domke reported on the adsorption of trimesic acid on Cu in ethanol. Cyclic voltammetric and Raman spectroscopic data indicate that a 2D adsorbate layer is formed at 0 V vs Cu. At higher coverages a multilayer of flat lying molecules is formed. When the potential is made more positive, a 3D MOF is produced due to the oxidation of the copper surface, as expected.^[33]

This paper reported the synthesis of MOF based on Cu and H₃BTC by potentiodynamic method, with emphasis on the effect of the solvent's nature on the morphology, structure and specific surface area of the obtained material. The CO₂ adsorption performance of the obtained MOF was also assessed.

2. MATERIALS AND METHODS

2.1. MOF synthesis procedure

As the E_{OCP} of Cu electrode in the electrolyte was about 0 V/SCE and the maximum potential of Autolab device is 10 V, the Cu-BTC material was synthesized by the potentiodynamic method from 0-10 V/SCE and a scan rate of 5 mV/s in an electrochemical cell with 3 electrodes. The 3-electrode cell includes a saturated calomel reference electrode (SCE), a working electrode with a limited area of 10 cm² and a counter electrode (Cu 99.61 %, 20×50×3 mm in size). The Cu electrode surface was treated with abrasive paper, rinsed with distilled water and solvent before each measurement. Electrochemical experiments were performed on an Autolab device.

Two different types of 80 mL solution were applied parallelly for the synthesis: (SS1) methanol + 0.05 M H₃BTC + 0.05 M NaNO₃ and (SS2) ethanol + 0.05 M H₃BTC + 0.05 M NaNO₃. In which: H₃BTC (C₉H₆O₆), M = 210.14 g/mol, Merck, 95 % purity. Methanol, ethanol and NaNO₃ are pure chemical of China.

2.2. Analysis of MOF properties

The surface functional groups of the synthesized Cu-BTC were determined by a FT-IR NEXUS 6700 (Nicolet) from 400-3000 cm⁻¹. The morphology was examined by a SEM S4800 (Hitachi). The total and micropore specific surface areas were determined by the BET and t-plot methods using Micromeritics ASAP 2060. Data in the relative pressure (p/p_0) range from 0.025 to 0.30 were used to determine the surface area by the BET equation: $\frac{1}{X[(\frac{p}{p_0})^{-1}]} = \frac{1}{X_m C} +$

$\frac{C-1}{X_m C} (\frac{p}{p_0})$, as mentioned in previous studies.^[34-36]

Here, X represents the weight of nitrogen adsorbed at a given p/p_0 , X_m represents the volume of gas adsorbed at standard temperature and pressure, and C is constant. The graph of the BET equation was linear with a positive slope of 0.207 g/cm³ STP and a fitting coefficient of 0.999 was obtained in our measurements. From the XRD measurement (Siemens D5000 of Bruker - Germany), the distance between crystal planes d and the phase structure of Cu-BTC can be determined. While the 1D structure Cu-BTC has monoclinic structure,^[37] the 3D structure Cu-BTC has faced-centered cubic structure with $a = b = c$ and $\alpha = \beta = \gamma = 90^\circ$.^[10,38] From the Miller index (hkl) of the 3D structure Cu-BTC, the lattice parameter a can be determined following the equation: $d = \frac{a}{\sqrt{h^2+k^2+l^2}}$ (*).

2.3. CO₂ adsorption procedure

The CO₂ adsorption capacity of the sample was measured at standard conditions (25 °C, 100 % CO₂) and more realistic conditions (40 °C, 15 % CO₂ in N₂) using a Macro-thermogravimetric system, designed by the Centre de Coopération Internationale en Recherche Agronomique pour le Développement (CIRAD) and the University of Science and Technology of Hanoi (USTH) (figure 1).

The reactor consists of a 7.5-cm ceramic tube (1)

placed inside a Carbolite electrical furnace (2). The atmosphere can be generated by injecting desired gases (CO₂ or N₂) through a 2m-coiled pre-heater (3). For each experiment, the sample placed in the holder (4) is lifted to the center of the reactor. The mass loss is continuously weighed by a weight scale (5) and recorded to a computer. Meanwhile, an extractor (6) extracts the flue gas out of the system. The dried sample was exposed under a CO₂ gas flow of 3 L/min

at different concentrations. When a stabilized uptake mass was obtained, the CO₂ flow was switched to pure N₂ to regenerate the sample. The CO₂ adsorption performance calculated as follows:

$$P_{CO_2}(\%) = \frac{m_f - m_i}{m_i} \times 100$$

where m_i and m_f are the initial and final mass of the material before and after adsorption, respectively.

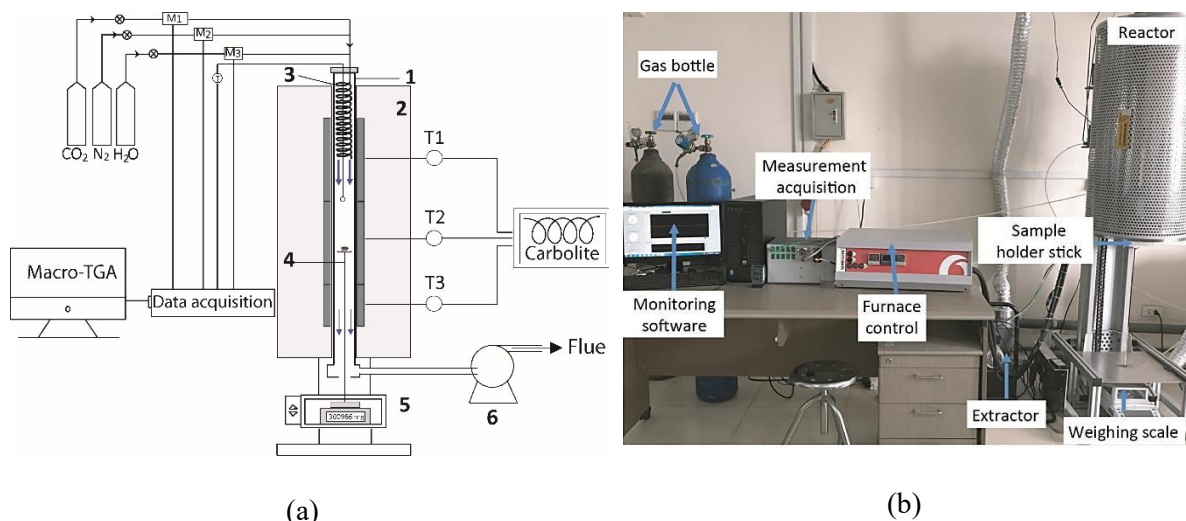


Figure 1: The macro-thermogravimetric reactor: (a) diagram and (b) real setup

3. RESULTS AND DISCUSSION

3.1. Characteristics of the synthesized material

Figure 2 presents the value of the current when varying the potential from 0 to 10 V/SCE (scan rate 5 mV/s) at the Cu working electrode in SS1 and SS2 solutions. In ethanol, the response current was very low. At $E = 10$ V/SCE, the anodic current intensity was 0.05 A. In methanol, the current was markedly increased with the potential and at $E = 10$ V/SCE, an anodic current of 0.35 A was reached. A higher anodic current generally results in a faster process and this is thus expected to be more favorable for the production of the material. This result can be explained by the higher polarization - expressed by the dielectric coefficient - of methanol (having a value of 33) compared to that of ethanol (having a value of 24.55). Moreover, the structure of ethanol molecules is more voluminous than that of methanol. The -C₂H₅ group is bigger and thus can push electrons stronger than the -CH₃ group, leading to the higher polarization of O-H bond in methanol than in ethanol. Therefore, the solvation ability of NaNO₃ in methanol is much better compared to that in ethanol, leading a higher conductivity of the solution. The IR spectra of the H₃BTC and the synthesized Cu-BTCs in methanol and ethanol are shown in figure 3. For H₃BTC, the

bands at 1710, 1180 and 1275 cm⁻¹ could be attributed to the C=O, C-O stretching vibration and the bending vibration of O-H of the carboxylic acid group. The symmetric and antisymmetric stretching of O-C=O group was observed in the range 1380-1620 cm⁻¹. The broad band observed in the 2500-3300 cm⁻¹ region corresponded to the O-H stretching of the carboxylic group. For Cu-BTC, the peaks at 1709, 1183, 1245 cm⁻¹ and in the range 1376-1630 cm⁻¹ could be attributed to the stretching vibrations $\nu_{C=O}$ and ν_{C-O} , the bending vibration of O-H, the symmetric and the antisymmetric stretching of O-C=O group, respectively. The broad band corresponding to the O-H stretching of the carboxylic groups is shifted to the 3100-3600 cm⁻¹ region, indicating the presence of loosely bound water molecules in Cu-BTC (table 1).^[5,17,38-40] From the high intensity of peak at 1275 cm⁻¹ in H₃BTC (corresponding to the bending vibration of O-H of the carboxylic acid group) to the lower intensity in Cu-BTC synthesized in SS1 (at 1245 cm⁻¹) and we cannot observe this peak in SS2, it can be concluded that the sample obtained from SS1 still have acid groups and in SS2 all H₃BTC was fully deprotonated. A similar phenomenon is also observed in the broad band from 2500-3300 cm⁻¹ or peak at 2553 cm⁻¹ (O-H stretching of the carboxylic group). This result will be confirmed by the XRD analysis in the next part.

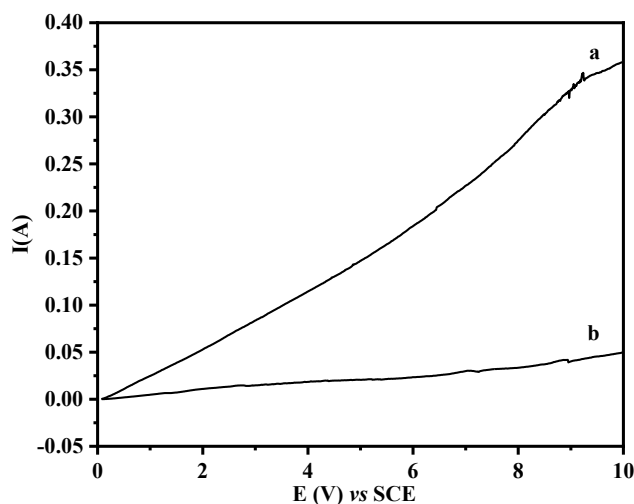


Figure 2: Current as a function of potential for the potentiodynamic synthesis process ($v = 5 \text{ mV/s}$) of Cu-BTC in SS1 (a) and SS2 (b)

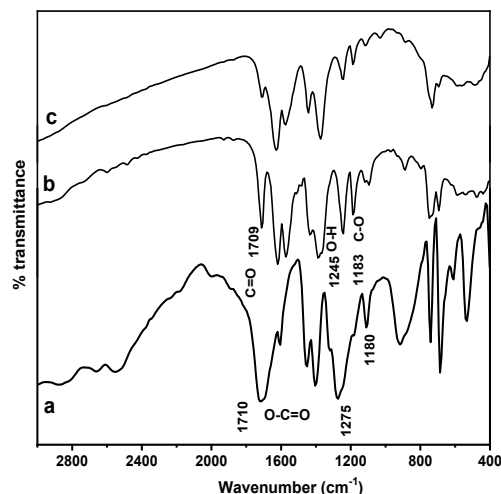


Figure 3: IR spectra of H₃BTC (a) and Cu-BTC synthesized by the potentiodynamic method in SS1 (b) and SS2 (c)

Table 1: Vibration groups of H₃BTC and CuBTC (wavenumbers are expressed in cm^{-1}).

Vibration group	VC=O	VC-O	VO-H bonding	VO-C=O	VO-H stretching
H ₃ BTC	1710	1180	1275	1380-1620	2553 (2500-3300)
Cu-BTC ^[17]	1710	1193	1232	1380-1620	
Cu-BTC synthesized in SS1	1709	1183	1245	1376-1630	3100-3600
Cu-BTC synthesized in SS2	1713	1188		1376-1630	3100-3600

The phase structure of the Cu-BTCs synthesized in SS1 and SS2 was shown in figure 4. Comparing the results with data from the literature (table 2),^[20,41] it can be concluded that the Cu-BTC synthesized in SS1 is a mixture of 1D structure catena-triaqua μ -[1,3,5-benzenetricarboxylato]-copper(II) corresponding to the formula $[\text{Cu}(\text{BTC})(\text{H}_2\text{O})_3]_n$ and micropillar structure $\text{Cu}(\text{BTC}) \cdot 3\text{H}_2\text{O}$.^[20,37,41] This will be confirmed by the SEM result below. The characteristic peaks of the 1D structure Cu-BTC were displayed at 2θ : 9.2°; 9.9°; 10.9° and 13.5°. Meanwhile, the Cu-BTC synthesized in SS2 has a 3D structure with 2θ : 6.7°; 9.5°; 11.6°; 13.5°. It is known that the formation

of 1D or 3D structure Cu-BTC depends on the deprotonation ability of H₃BTC. If the deprotonation is complete, Cu^{2+} ions can react with the 3 carboxylate groups, and the 3D structure Cu-BTC can be obtained. In the case where only partial deprotonation occurs, Cu^{2+} ions only react with two instead of three $-\text{COO}^-$ groups and a zigzag structure (1D) is observed.^[20] This result can be explained by the fact that C_2H_5^+ can push electrons to O-H stronger than CH_3^+ (or the Inductive effect +I of C_2H_5^+ is stronger than CH_3^+) leading to the nucleophilic property of OH^- in ethanol is more than OH^- in methanol. This property makes H₃BTC deprotonate totally in ethanol.

Table 2: The position of diffraction peaks of CuBTC synthesized by the potentiodynamic method in SS1 and SS2 compared to the CuBTC from the literature

Cu(BTC)(H ₂ O) ₃ 1D ^[20]	9.3	10.0	11.0	13.4	17.1	18.6	21.4	22.1	24.5					
Cu(BTC)·3H ₂ O [41]			11.4	14.0	16.4	17.1	18.8	19.5	21.4	21.7	22.9	24.6		
SS1	9.2	9.9	10.9	11.31	13.5	16.13	17.0	18.6	19.46	21.3	22.0	22.8	24.3	
$\text{Cu}_3(\text{BTC})_2$ ^[20]	6.8	9.6	11.8	13.6	14.9	15.2	16.7	17.7	19.2	20.5	21.5	23.1	23.5	24.2
SS2	6.7	9.5	11.6	13.5	14.6	15.0	16.5	17.5	19.0	20.2	21.2	23.3	24.1	

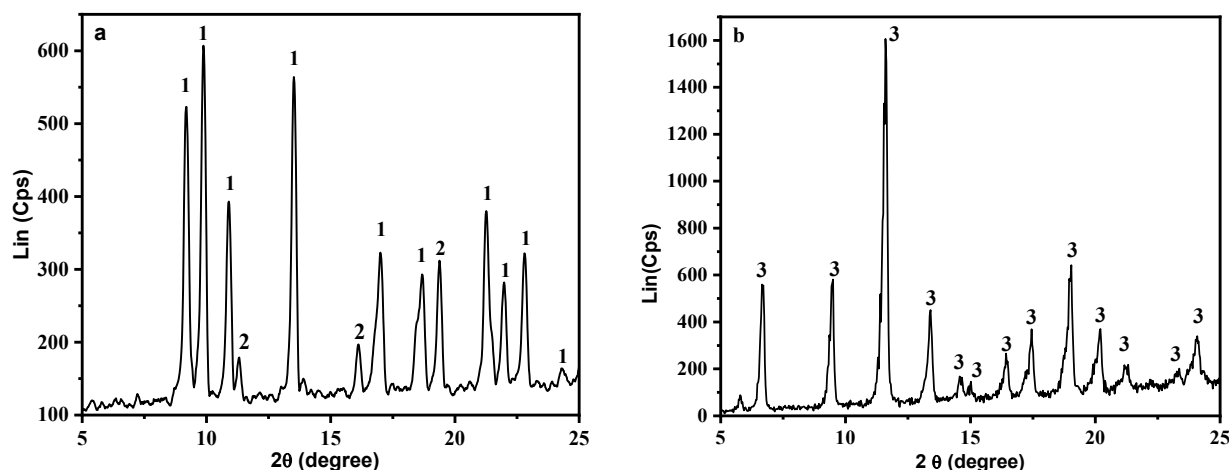


Figure 4: Xray diffraction patterns of Cu-BTC synthesized potentiodynamically in SS1 (a) and SS2 (b). Characteristic peaks: 1 = $[\text{CuBTC}(\text{H}_2\text{O})_3]_n$ 1D, 2 = $\text{CuBTC} \cdot 3\text{H}_2\text{O}$, 3 = $\text{Cu}_3(\text{BTC})_2$ 3D

From the 2θ value, (hkl), the distance between crystal plane d , the lattice parameter $a = 26.454 \text{ \AA}$ of Cu-BTC 3D was calculated, which is similar with the previous result of Wojciech Starosta *et al.* (26.346 \AA).^[38]

Figure 5 presents the SEM images of the Cu-BTC synthesized in SS1 (a) and SS2 (b). In SS1

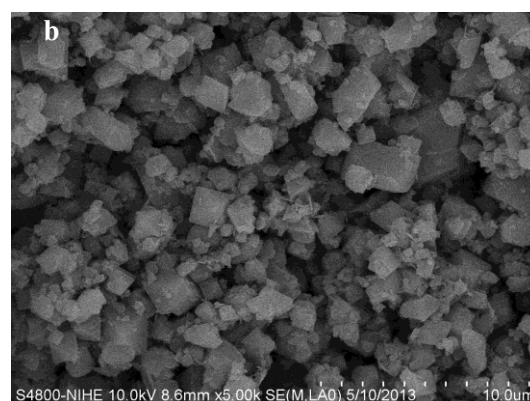
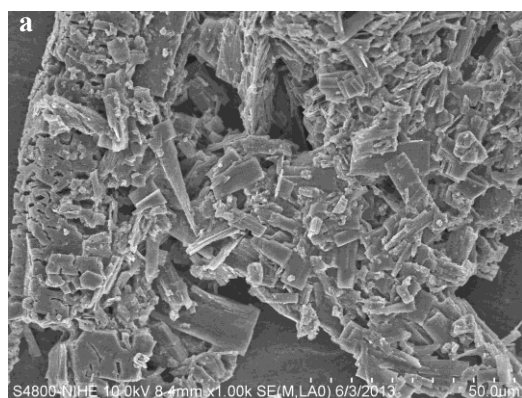


Figure 5: SEM images of Cu-BTC synthesized in SS1 (a) and SS2 (b)

Figure 6 presents the adsorption isotherms in the range p/p_0 from 0 to 0.99 at 77 K. While the 1D structure Cu-BTC's isotherms followed type I in the IUPAC classification, which indicated a non-porous structure, the isotherms of the 3D structure Cu-BTC were much closer to a type IV, indicating a highly porous material. A large-range hysteresis loop was also observed, suggesting the presence of bottleneck pores in the 3D structure Cu-BTC. This structure is well suited for the easy capture of CO_2 molecules during adsorption. As expected, the total specific surface area and total pore volume, determined from the BET method, of the 1D structure Cu-BTC were much smaller than that of the 3D structure Cu-BTC (table 3). The micropore volume of the 1D and 3D

with methanol, the obtained Cu-BTC crystals have a size that varies in a large range, from small particles with a size of about 1 μm to a plate shape of about 20 μm . The image is satisfied with the XRD data. In the case of SS2 with ethanol, the Cu-BTC has a cubic shape with sizes ranging from 50 to 500 nm, suitable with the XRD data above.

structure Cu-BTC was 0.01 cm^3/g and 0.25 cm^3/g , respectively. It can be confirmed that the 3D structure Cu-BTC is mainly made of porous structures, which is favorable for CO_2 adsorption.^[4,42]

3.2. CO_2 adsorption capacity

Based on the results of the properties of the two synthesized materials, the 3D-structure Cu-BTC was selected for further CO_2 adsorption experiments. The optimum CO_2 uptake (expressed in mass percentage) of material (200 mg) was determined using the Macro-TGA at 25°C and 100% CO_2 (figure 7). It can be observed that the maximum CO_2 uptake reached 3.6 mmol/g which was calculated by the following

formula:

$$Q_{max} = \frac{\text{uptake} \times 1000}{M_{CO_2}} = \frac{15.8\% \times 1000}{44} = 3.6 \text{ (mmol.g}^{-1}\text{)}$$

The presence of micropores in the 3D-structure

Cu-BTC could be responsible for this CO₂ uptake. Previous research also highlighted the importance of the specific surface area of the material, especially the microporous structures, on the CO₂ adsorption capacity of the material.^[43,44]

Table 3: Specific surface area and pore volume of CuBTC

Material	S_{Total} (m ² /g)	V_{Total} (cm ³ /g)	V_{Micro} (cm ³ /g)	$d_{Mean\ pore}$ (nm)
1D Cu-BTC	21	0.01	0.01	78.22
3D Cu-BTC	620	0.52	0.25	3.37

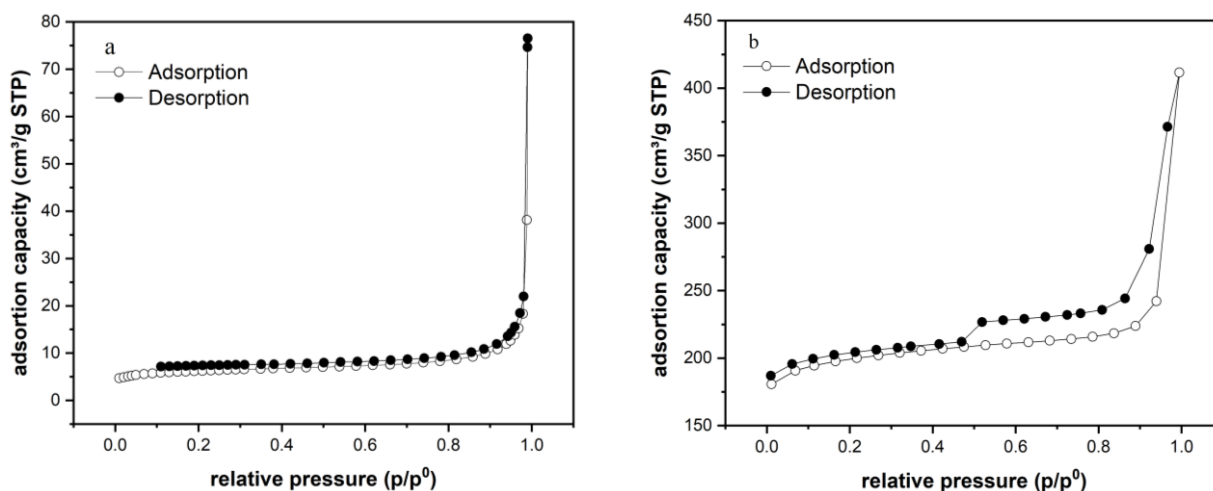


Figure 6: Adsorption isotherms of Cu-BTC synthesized in SS1 (a) and SS2 (b)

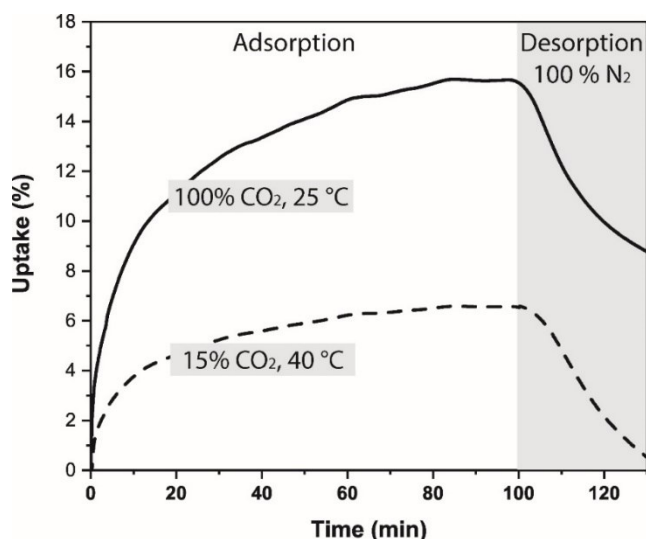


Figure 7: CO₂ uptake of 3D-structure Cu-BTC

To better evaluate the CO₂ adsorption performance of the 3D-structure Cu-BTC, the above result was compared with those of other common CO₂ adsorption materials, such as biomass-derived activated carbons, MOFs, and zeolites (table 4). Most of the CO₂ uptakes of common CO₂ adsorption materials are found in between 2 and 4 mmol/g. Thus,

the 3D-structure Cu-BTC synthesized in our research reached a good adsorption capacity, which is in the upper range when comparing with some other materials. It should also be noted that some materials are made from quite complex activation processes and are usually quite time-consuming. This makes it relatively difficult to upcle the production process. These comparative results imply that the use of a simple and quick process as potentiodynamic method to produce the 3D-structure Cu-BTC brings some unique advantages in practical conditions.

The CO₂ uptake of the 3D-structure Cu-BTC was also tested in closer practical conditions, taking into account the diffusion limitation within the bed of sorbent, higher working temperature, and low CO₂ concentration in flue gases. Hence, a bed of 2-cm material (about 2g in mass) was tested for CO₂ adsorption at 40 °C and 15% CO₂ concentration (in N₂). The result (figure 7) showed a significant decrease in the amount of CO₂ uptake, at only 1.3 mmol/g. It can be explained by the fact that high temperatures and low CO₂ concentrations are not favored for physical adsorption is the main adsorption mechanism. Nevertheless, the CO₂ uptake is still reasonable even at high temperatures, and can be

comparable with common adsorbents used in the industry, e.g. hydrotalcites.^[49] A small test for cyclic adsorption/desorption was also performed for the 3D-structure Cu-BTC at 25 °C and 100 % CO₂. When a maximum mass was observed, the CO₂ flow was

stopped and the N₂ flow was established to regenerate the sample. This protocol was repeated for 5 times. The results showed that the material still achieved a stable CO₂ uptake, suggesting a promising possibility to reuse the material.

Table 4: CO₂ uptake of CuBTC and some other materials

Group	Material	Production condition	CO ₂ uptake (mmol g ⁻¹)
	3D CuBTC (this study)	Synthesized by potentiodynamic procedure	3.60
	Pomegranate peels ^[43]		4.11
	Carrot peels ^[43]	Activated by KOH, ratio 1:1 at 700 °C	4.18
	Fern leaves ^[43]		4.12
		Activated by KOH, ratio 1:5 at 600 °C	1.12
	Empty fruit punch ^[44]	Activated by KOH, ratio 1:5 at 800 °C	2.63
Biochar/ Activated carbon		Hydrothermal at 150, 250, and 350 °C then activated by KOH, ratio 1:5 at 800 °C	3.40 – 3.71 – 2.18
	Olive stone ^[45]	Activated by CO ₂ at 800 °C during 1h, 2h, 4h, 6h and 8h	2.5 – 2.7 – 2.9 – 3.1 – 3.1
		Pyrolyzed at 600 °C during 4h, then activated by KOH at 800 °C with ratios 1:1, 1:2, 1:3, and 1:4	3.36 – 3.82 – 3.48 – 3.73
	Coffee Ground ^[46]	Pyrolyzed at 600 °C during 4h then activated by KOH, ratio 1:2 at 700, 800 and 900 °C	2.42 – 3.82 – 2.62
	Zeolite 13X ^[47]		0.36
	Zeolite 13X-MEA ^[47]		0.45
	Zeolite Meso-13X-PEI _{800MW} ^[47]		1.32
Zeolite	Zeolite 13X-PEI _{600000MW} ^[47]	Physical impregnation of amine (PEI) on different zeolites	1.22
	Zeolite SBA-PEI _{750000MW} ^[47]		2.15
	Zeolite MCM41-PEI ^[47]		2.55
	MOF UiO-66-NH ₂ ^[48]		3.0
	MOF Cu-BTT _r ^[48]		3.2
	UiO-66-AD ₈ ^[48]		3.3
Metal-organic framework	IRMOF-74-III-CH ₂ NH ₂ ^[48]	Different MOF chemically synthesized	3.3
	SNU-100-Li _j ^[48]		3.4
	SUMOF-3 ^[48]		3.4
	LIFM-33 ^[48]		3.4

4. CONCLUSION

The Cu-BTCs were successfully synthesized by the potentiodynamic method, from 0 to 10 V/SCE at a scan rate of 5 mV/s in the presence of air. The XRD results indicated that the mixture of 1D structure

[Cu(BTC)(H₂O)₃]_n and micropillar structure Cu(BTC).3H₂O was obtained when using methanol as a solvent, while the cubic 3D structure Cu-BTC was obtained when using ethanol. With a large total specific surface area obtained, the CuBTC 3D mainly consisted of microporous structures, suitable for CO₂

adsorption. The tests with CO₂ adsorption, both at standard conditions and more realistic conditions confirmed that the 3D structure Cu-BTC could be a potential candidate for CO₂ capture technologies.

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Conflict of interest. *The authors declare no competing interests.*

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